

PROBLEM 1 LEVEL 1

a) The liquid A-B mixture is NOT ideal since $\Delta H_{mix} \neq 0$

b) For the liquid :

$$\begin{cases} \mu_A^l = \mu_A^{\circ l} + RT \ln a_A \\ \mu_B^l = \mu_B^{\circ l} + RT \ln a_B \end{cases}$$

For the gas phase :

$$\begin{cases} \mu_A^g = \mu_A^{\circ g} + RT \ln P_A \\ \mu_B^g = \mu_B^{\circ g} + RT \ln P_B \end{cases}$$

* When we have pure A, then $a_A = 1$ $P_A = P_A^\circ =$ vapor pressure above pure A

$$\mu_A^l = \mu_A^{\circ l} = \mu_A^{\circ g} + RT \ln P_A^\circ$$

or $RT \ln P_A^\circ = \mu_A^{\circ l} - \mu_A^{\circ g}$

Similarly for B :

$$RT \ln P_B^\circ = \mu_B^{\circ l} - \mu_B^{\circ g}$$

* When $x_B = 0.95$, then

$$\mu_A^l = \mu_A^{\circ l} + RT \ln a_A = \mu_A^{\circ g} + RT \ln P_A$$

$$RT \ln a_A = \underbrace{-(\mu_A^{\circ l} - \mu_A^{\circ g})}_{-RT \ln P_A^\circ} + RT \ln P_A$$

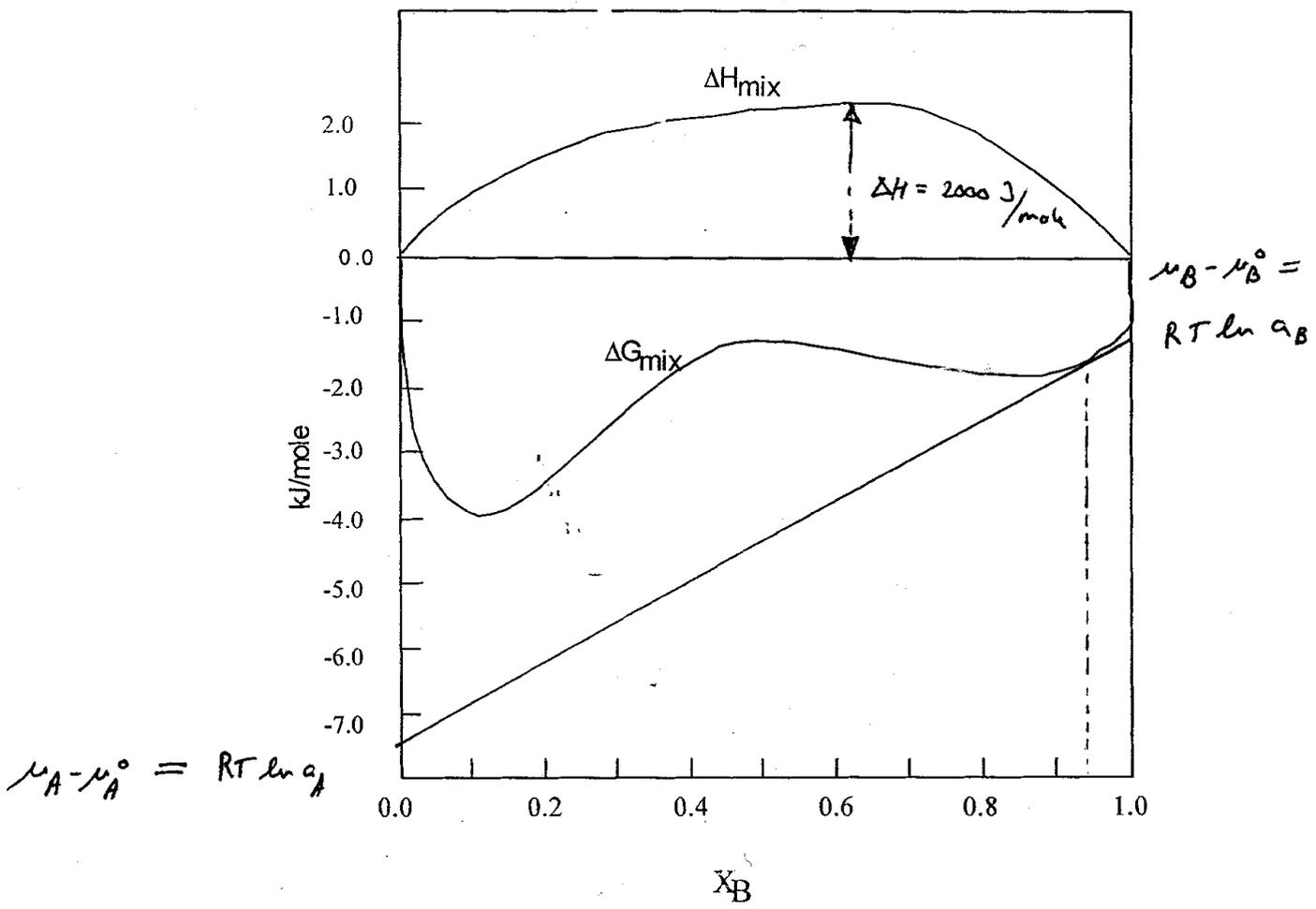
$$RT \ln a_A = RT \ln \left(\frac{P_A}{P_A^\circ} \right)$$

PROBLEM 1 CONTINUED

Similarly for B

$$RT \ln a_B = RT \ln (P_B/P_B^\circ)$$

$RT \ln a_A$ and $RT \ln a_B$ can be read off of the plot for ΔG_{mix} using the intercept rule.



From the plot, we can estimate $a_A = 0.41$ and $a_B = 0.83$

$$P_A = a_A P_A^\circ = 4.1 \times 10^{-6} \text{ atm}$$

$$P_B = a_B P_B^\circ = 8.3 \times 10^{-3} \text{ atm}$$

PROBLEM 1 CONTINUED

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c) Total # of moles of mixture: $N_A + N_B = 2.5$ moles

$$x_A = 0.4$$

$$x_B = 0.6$$

at $x_B = 0.6$ $\Delta H_{mix} \approx 2000$ J/mole (see figure)

$\Delta H_{mix} > 0$, so mixing will draw heat from the environment

amount of heat sucked in:

$$Q = \underbrace{2.5}_{N_{tot}} (2000) = 5000 \text{ J}$$

PROBLEM 2 (LEVEL 1)

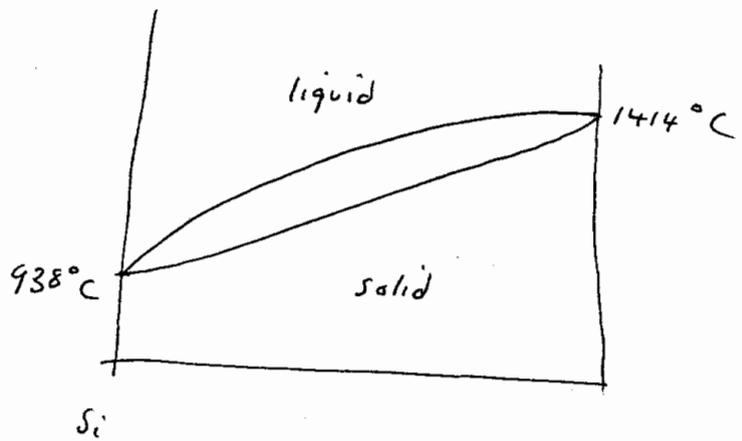
microscopic mechanisms that contribute to the increase in entropy of ferro magnetic Fe-75% Ni as T is increased:

- (i) magnetic disordering of the magnetic moments
- (ii) configurational disordering of the Ni and Fe atoms
- (iii) entropy arising from lattice vibrations

PROBLEM 3 (LEVEL 1)

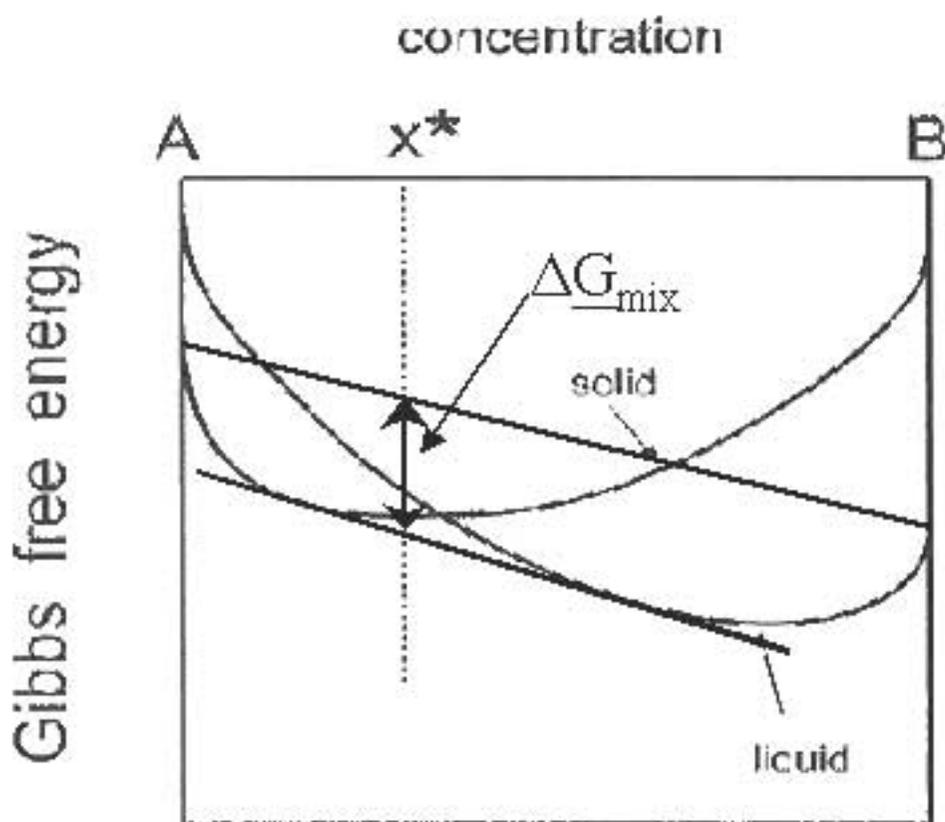
Si and Ge form ideal solutions in both liquid and solid state.

⇒ phase diagram at high T is a lens diagram



When Ge is added to Si, the melting point of the solution will therefore be higher than 938°C

Problem 4 (Level 1)



PROBLEM 1 (LEVEL 2)

Cu & Au can form a ~~solid~~ solid solution at high T or an ordered compound at low T

a) * mix ~~one~~ 1 mole of Cu and 1 mol of Au at 1150° K

* assume ideal solution

$$\Rightarrow \Delta G_{mix} = RT [x \ln x + (1-x) \ln (1-x)]$$

Where x is the Au concentration and is = 0.5

$$\Delta G_{mix} = -6627 \text{ J/mol at } T = 1150^\circ \text{ K}$$

* Since 1 mole of Cu and 1 mole of Au \Rightarrow 2 moles of mixture

$$\Delta G = 2 \times \Delta G_{mix} = -13254 \text{ J}$$

b) For ordered compound,

$$\Delta G_{Cu_{0.5}Au_{0.5}} = \Delta H_{Cu_{0.5}Au_{0.5}} \quad (\text{since } \Delta S_{CuAu} = 0)$$

$$\Delta G_{Cu_{0.5}Au_{0.5}} = -\frac{11904}{2} \text{ J/mol}$$

at the (first-order) order-disorder transition,

$$\Delta G_{Cu_{0.5}Au_{0.5}} = \Delta G_{mix}$$

This occurs at $T = 1033 \text{ K}$

PROBLEM 1 (LEVEL 2) CONTINUED

c) In ΔG_{mix} for the solid solution, we neglected ΔH_{mix} . Since this system has ordering tendencies, $\Delta H_{\text{mix}} < 0$. ~~This will depend~~ Inclusion of a negative ΔH_{mix} ~~in~~ in part (b) will decrease the calculated transition temperature.

PROBLEM 2

The ideal solution entropy formula is a better approximation at high temperature.

For the ideal solution entropy formula, it is assumed that every arrangement of A and B atoms are equally probable. In reality, this is not true since the probability of an arrangement ν is

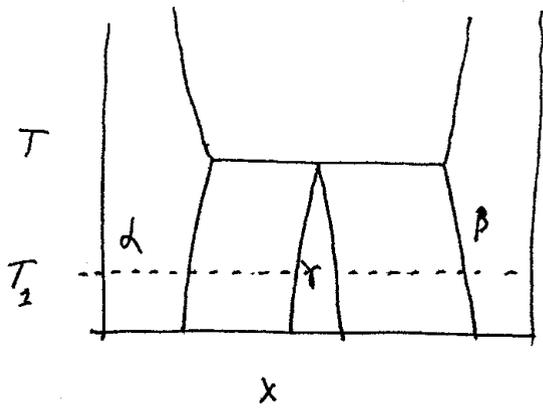
$$P_{\nu} \sim \exp\left(\frac{-E_{\nu}}{kT}\right)$$

Since different arrangements ν and η typically have different energies,

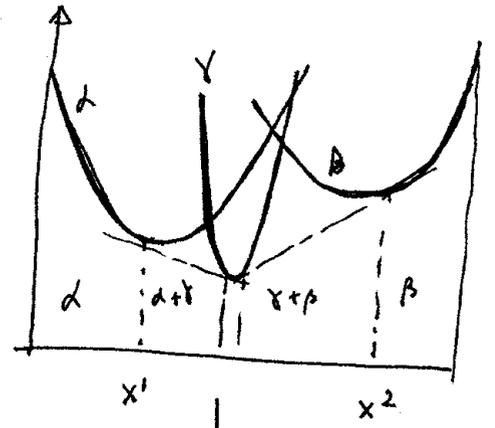
$$\Rightarrow \frac{P_{\nu}}{P_{\eta}} = \exp\left(\frac{-(E_{\nu} - E_{\eta})}{kT}\right) \neq 1$$

This ratio does, however, approach 1 as T is increased.

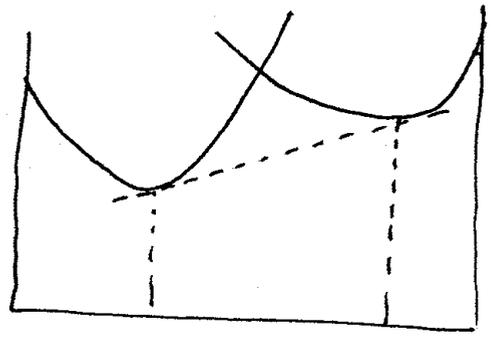
PROBLEM 3 (LEVEL 2)



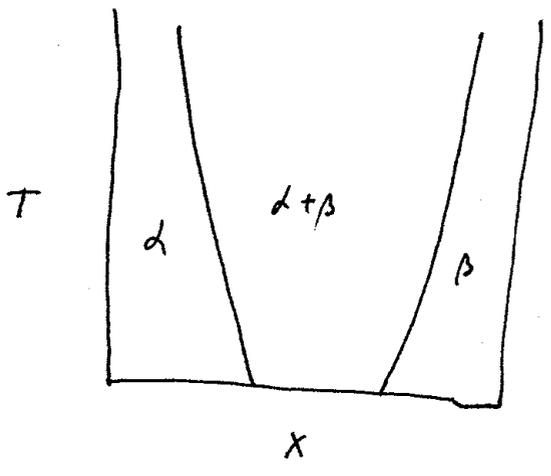
Free energies
at T_2



Suppression of the
 γ -phase yields



meta stable
phase diagram
(with γ suppressed)



PROBLEM 4 (LEVEL 2)

REGULAR SOLUTION MODEL

$$\Delta G_{mix} = ZW X_A X_B + RT [X_A \ln X_A + X_B \ln X_B]$$

with $W = W_{AB} - \frac{1}{2}(W_{AA} + W_{BB})$

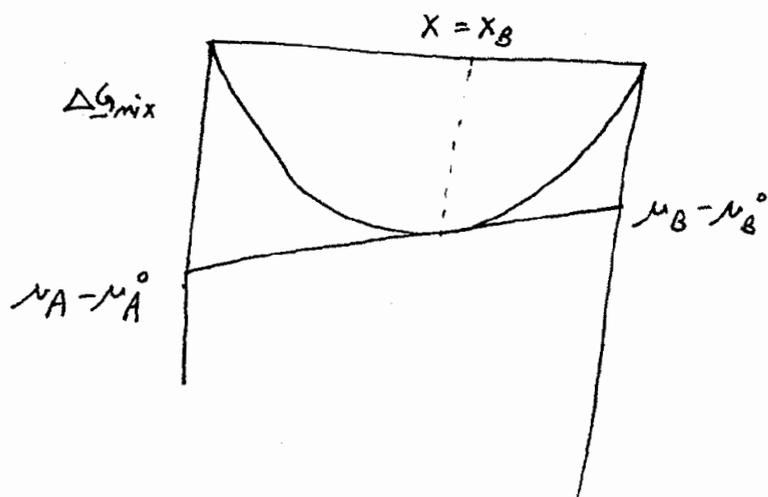
$$W = 630 \text{ J/mol}$$

$$Z = 8$$

a) (i) $\Delta H_{mix} = ZW X_A X_B$

(ii) $\Delta S_{mix} = -k [X_A \ln X_A + X_B \ln X_B]$

(iii) chemical potentials



$$\begin{cases} \mu_A - \mu_A^0 = \Delta G_{mix}(x) - \frac{d\Delta G_{mix}}{dx} x = ZW x^2 + RT \ln(1-x) \\ \mu_B - \mu_B^0 = \Delta G_{mix}(x) + \frac{d\Delta G_{mix}}{dx} (1-x) = \cancel{ZW(1-x)^2 + RT \ln x} \\ = ZW(1-x)^2 + RT \ln x \end{cases}$$

PROBLEM 4 (LEVEL 2)

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a) (iv) Activities

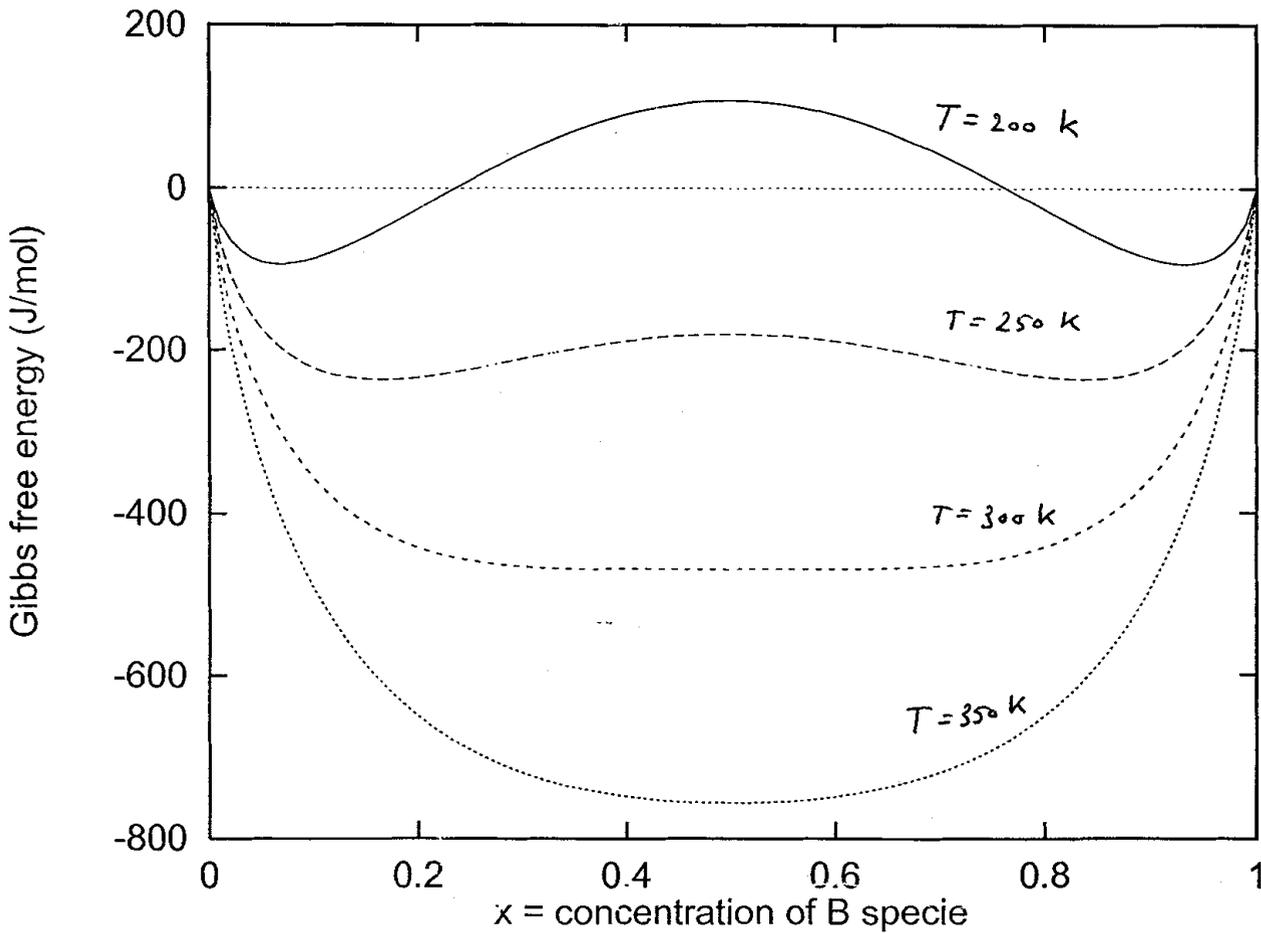
for B,

$$RT \ln a_B = zW(1-x)^2 + RT \ln x$$

$$\ln a_B = \frac{zW}{RT} (1-x)^2 + \ln x$$

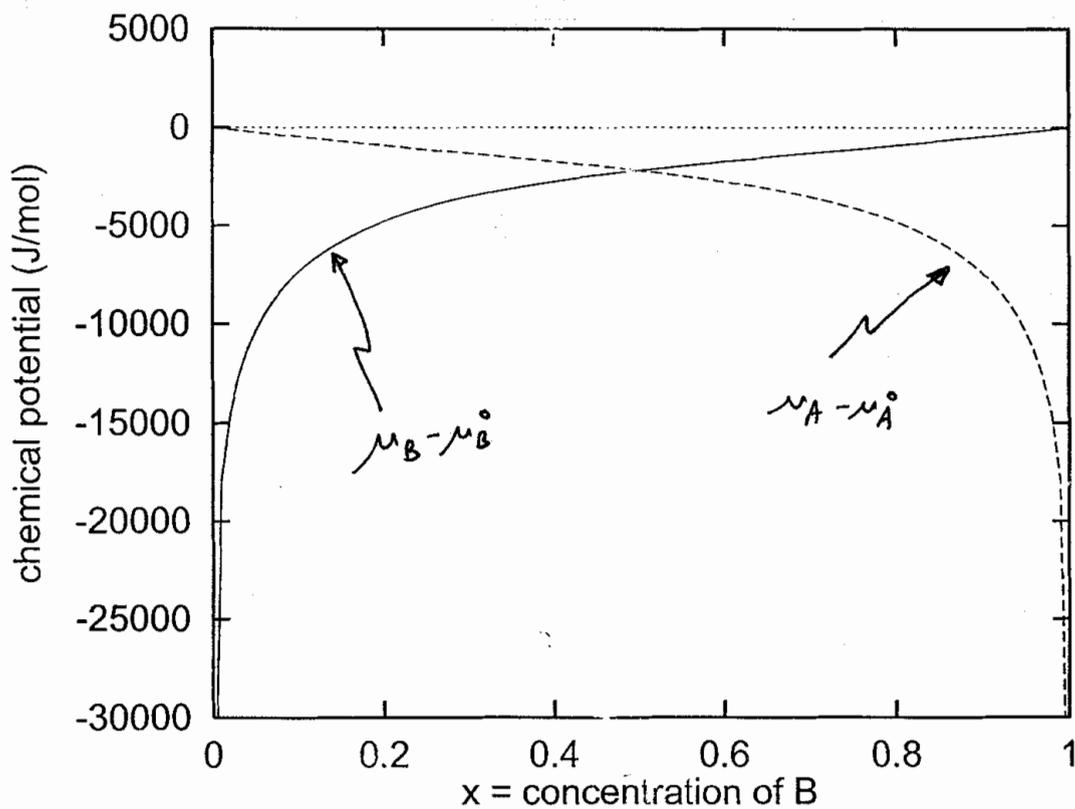
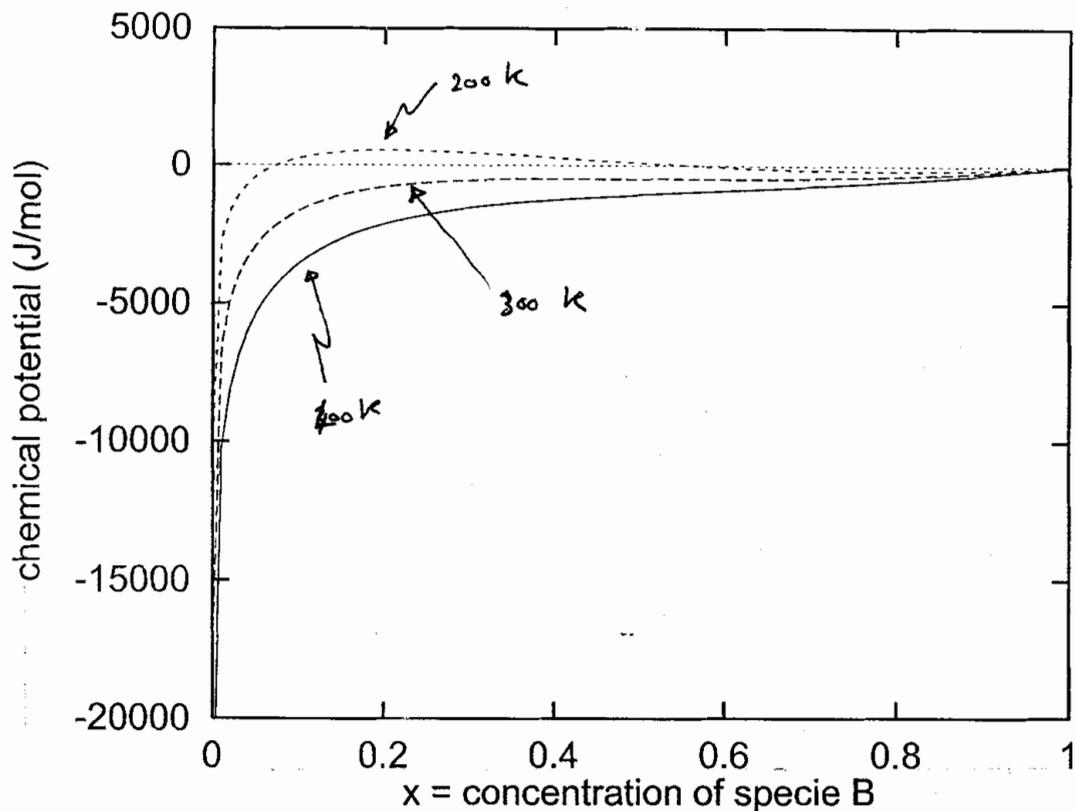
$$a_B = x \exp\left(\frac{zW}{RT} (1-x)^2\right)$$

b) ΔG_{mix} at different temperatures



g)

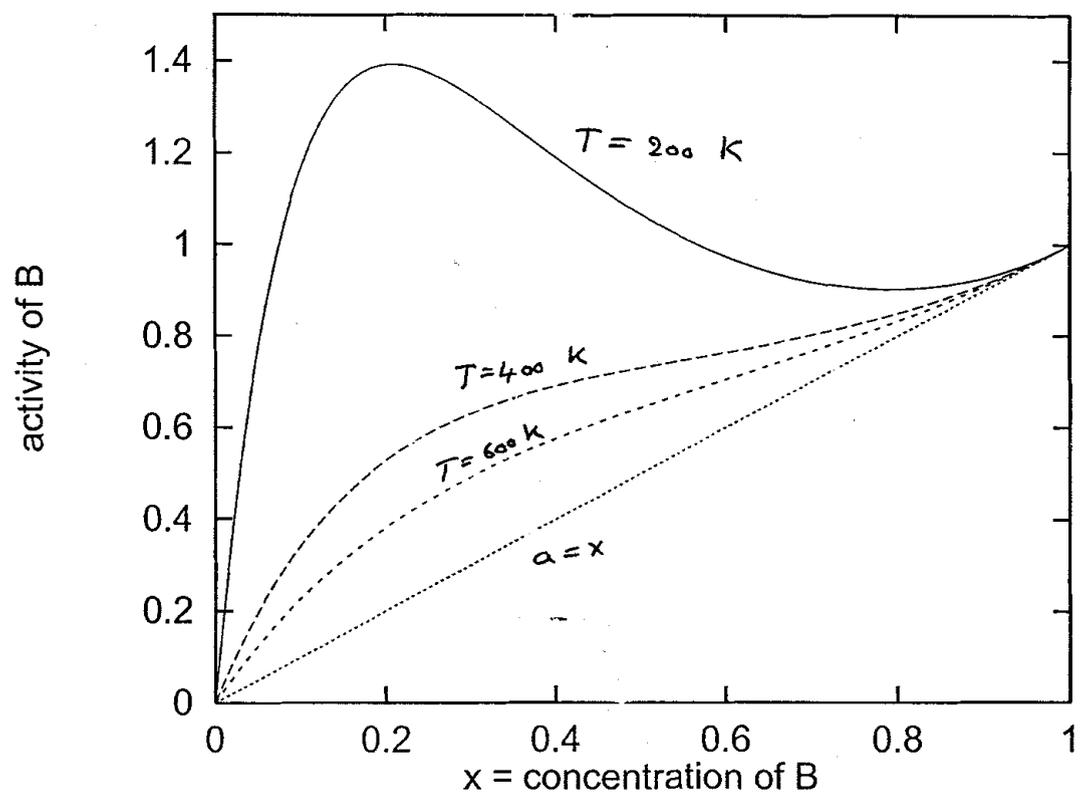
CHEMICAL POTENTIAL $\mu_B - \mu_B^\circ$



9)

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Activity of B



PROBLEM 5 (LEVEL 2)

	CO_2	+	H_2	=	CO	+	H_2O
initial state	0.2		0.5		0.2		0.1
final state	$0.2 - \lambda$		$0.5 - \lambda$		$0.2 + \lambda$		$0.1 + \lambda$

where λ = progress variable of the reaction

* Equilibrium constant of the reaction:

$$K = \frac{P_{\text{CO}} P_{\text{H}_2\text{O}}}{P_{\text{CO}_2} P_{\text{H}_2}} \quad \text{with} \quad P_i = \left(\frac{n_i}{n_{\text{total}}} \right) P$$

$$\Rightarrow K = \frac{(0.2 + \lambda)(0.1 + \lambda)}{(0.2 - \lambda)(0.5 - \lambda)} \quad (1)$$

* we also know that

$$K = \exp\left(\frac{-\Delta G^\circ}{RT}\right) \quad \text{with} \quad \Delta G^\circ = \mu_{\text{CO}_2}^\circ + \mu_{\text{H}_2}^\circ - \mu_{\text{CO}}^\circ - \mu_{\text{H}_2\text{O}}^\circ$$

as is given, $\Delta G^\circ = -2300$ cal at 1500K

$$\Rightarrow K = 2.16 \quad (2)$$

solving (1) and (2) yields

$$\lambda_{\text{eq}} = 0.117$$

in the final state we have:
8.3% CO_2 and 21.7% H_2O
38.3% H_2
31.7% CO

PROBLEM 6 (LEVEL 2)

- * Fe acts as a host for C atoms which occupy interstitial sites in the bcc structure of Fe.
- * Assume we have 1 mole of Fe in the bcc structure (i.e. N_A Fe atoms). Then we have $3N_A$ interstitial sites available for C.
- * Assume we take n C atoms from its pure reference state (e.g. graphite) and insert them into the Fe crystal. (we consider dilute C concentration such that $n < N_A$)

Then:

$$\Delta H = \frac{n}{N_A} E_i$$

$$\Delta S = k \ln \left[\frac{(3N_A)!}{(3N_A - n)! n!} \right] = k \left[\ln(3N_A)! - \ln(3N_A - n)! - \ln n! \right]$$

$$\stackrel{\text{Stirling}}{=} k \left[3N_A \ln \left(\frac{3N_A}{3N_A - n} \right) + n \ln \left(\frac{3N_A - n}{n} \right) \right]$$

$$\Delta G = \Delta H - T\Delta S$$

$$= \frac{n}{N_A} E_i + kT \left[3N_A \ln \left(\frac{3N_A - n}{3N_A} \right) + n \ln \left(\frac{n}{3N_A - n} \right) \right]$$

$$\text{let } y = \frac{n}{N_A}$$

PROBLEM 6 (LEVEL 2) CONTINUED

$$\Delta G = y E_i + RT \left[3 \ln \left(1 - \frac{y}{3} \right) + y \ln \left(\frac{y/3}{1 - y/3} \right) \right]$$

$$= y E_i + RT \left[3 \ln \left(1 - \frac{y}{3} \right) + y \ln \left(\frac{y}{3} \right) - y \ln \left(1 - \frac{y}{3} \right) \right]$$

$$\Delta G = y E_i + RT \left[3 \left(1 - \frac{y}{3} \right) \ln \left(1 - \frac{y}{3} \right) + y \ln \left(\frac{y}{3} \right) \right]$$

→ The free energy for $Fe C_y$

b) $\mu_c = \frac{\partial \Delta G}{\partial (n/N_A)}$ where $\frac{n}{N_A}$ is the number of moles of c in one mole of Fe

$$\mu_c = E_i + RT \ln \left(\frac{y/3}{1 - y/3} \right)$$